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Strain gradient enhanced crystal plasticity model: Application to electric steels

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M2i Cluster-1, project number: T18011

Industrial partner: Bosch Transmission Technology

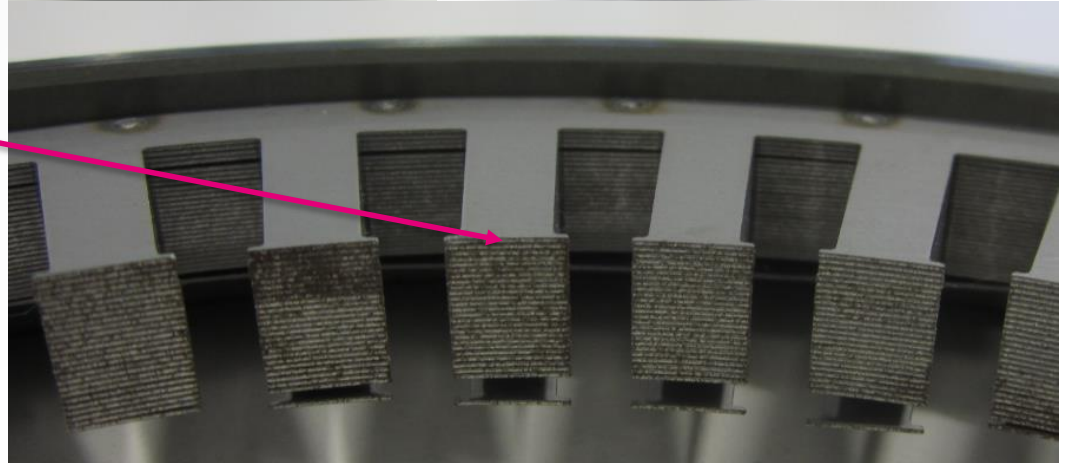
Supervisor: Dr. E.S. Perdahcioglu, University of Twente

Agenda

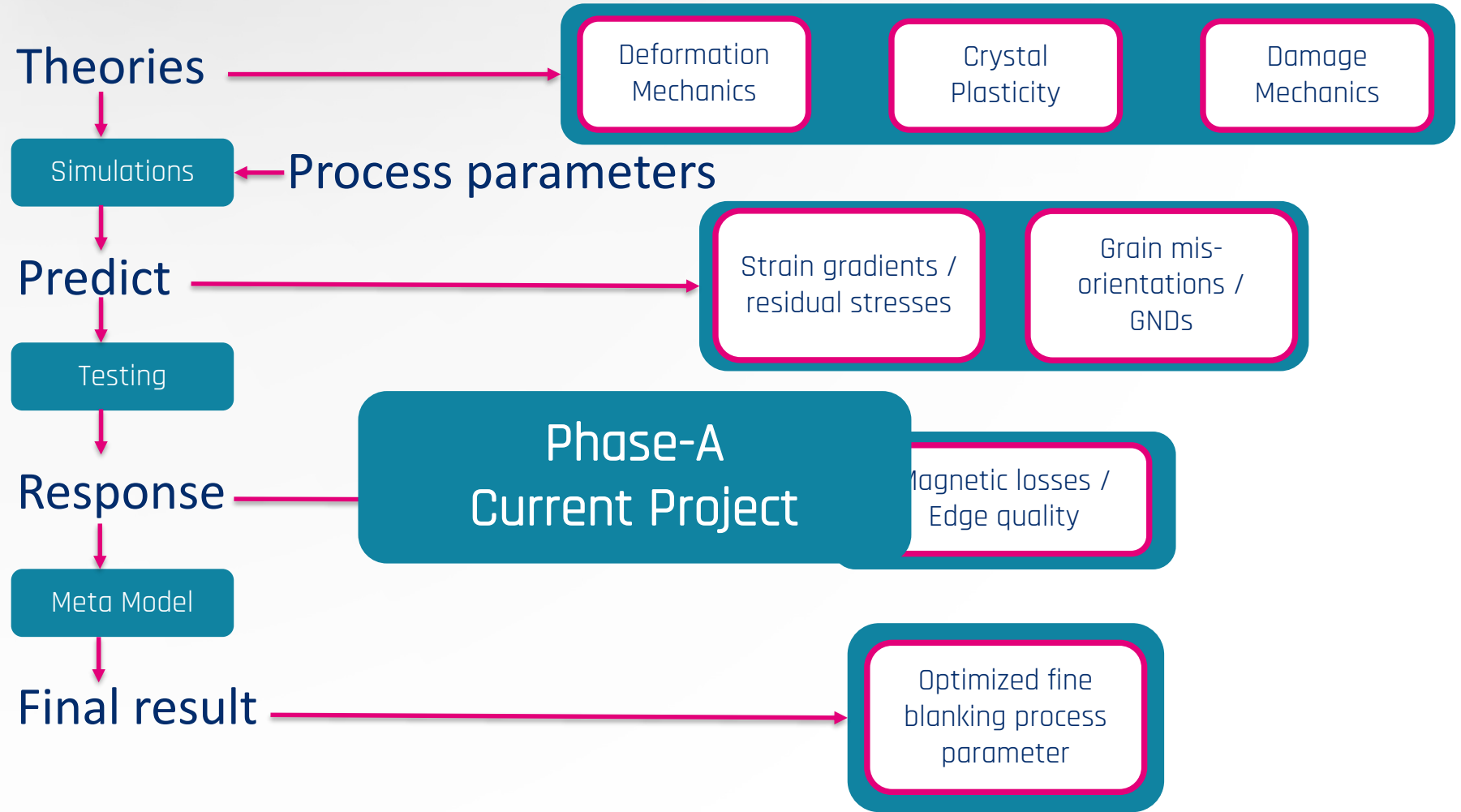
- Industrial Aim
- Scientific Approach and Challenges
- Introduction
 - Electric Steel
 - Strain gradients and grain orientation
 - GNDs and KAM
- Strain gradient enhanced crystal plasticity
- EBSD measurements
- Tensile tests and simulation results
- Fine blanking with crystal plasticity
- Conclusions

Industrial Aim

- Supply laminate stacked cores for e-motors
- Reduce magnetic loss for competitive edge
- Fine blanking is used for cutting laminates
- Grain orientation at the cutting edge influences the magnetic loss
- Optimize the fine blanking process parameters



Overall Scientific Approach



Scientific Challenges

Simulations with Gradient Enhanced Crystal Plasticity (physically based)

- RVE cannot be defined (Grains are too large)
- Grain size and orientation variations (response will vary)
- Re-meshing
- Large computation time

Simulations with Strain Gradient continuum Plasticity

- Relatively low computation time
- Grain size and orientation not taken into account
- Model parameters for accurate predictions? ←

Scientific Challenges (cont'd)

Tensile tests

- Rolling direction
- Transverse direction

Gradient Enhanced
Crystal plasticity
Model parameters

EBSD

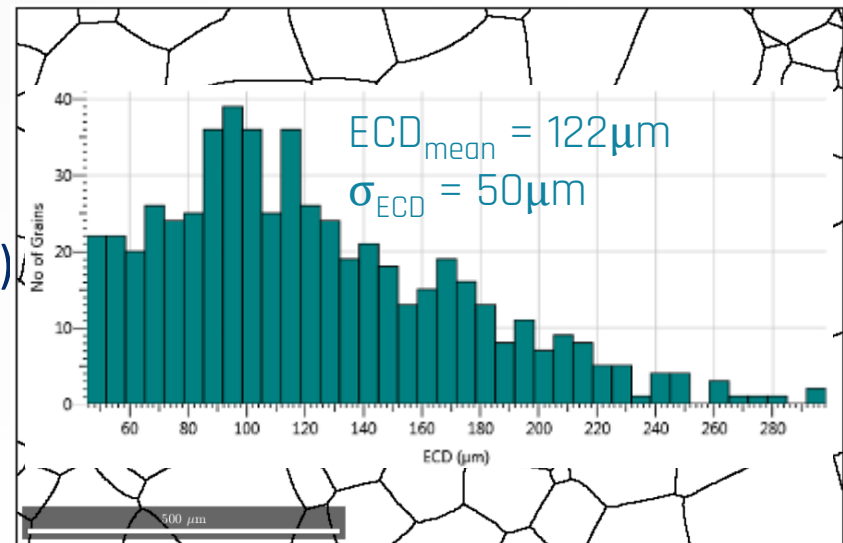
- Undeformed specimens
- Deformed specimens
- Grain size and orientations

Fit Model
parameters

- Simulations
- Actual microstructure
- Actual Grain data

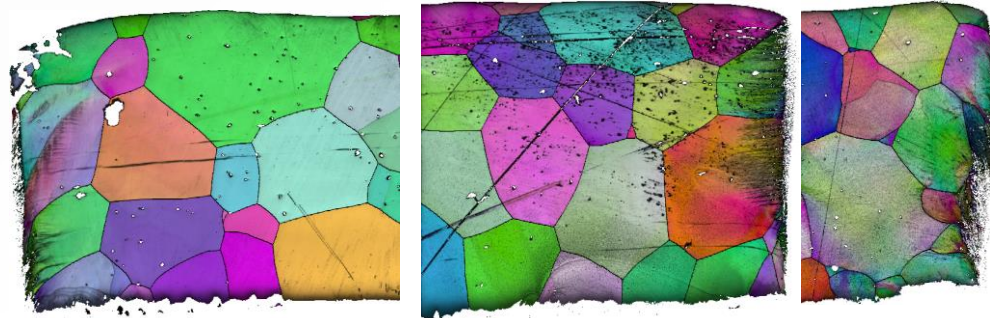
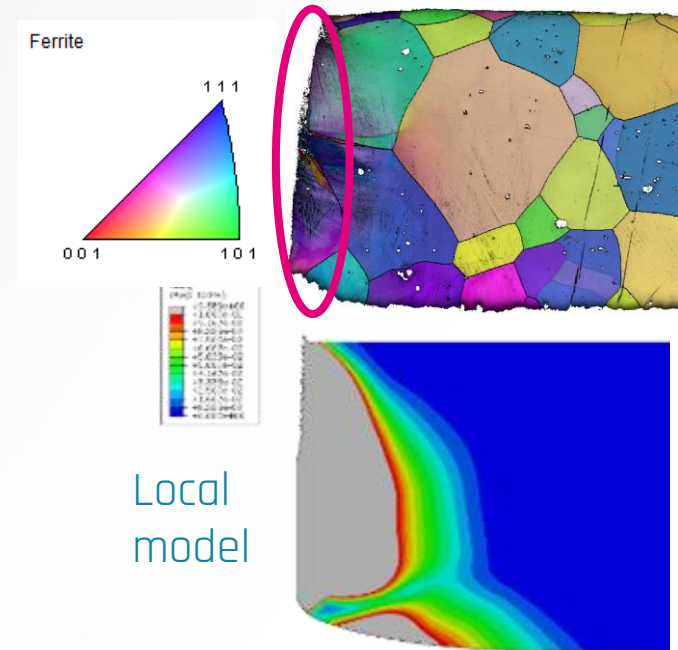
Introduction (Electric Steel)

- Excellent soft magnetic properties (magnetization behavior, permeability, coercive field and magnetic losses)
- Main Application: Electrical motors and transformers
- BCC crystals with $\langle 100 \rangle$ as most favorable magnetization direction
- In this project
 - Non-oriented electric steel (planar isotropic while $\langle 100 \rangle$ preferred normal to sheet plane)
 - 3.3% Si, 0.9% Al, $t=0.2\text{mm}$
 - Varnish coating C3 classification
 - Large grains



Introduction (Gradient + Grain orientations)

- Why gradient enhanced?
- Hardening is a function of strain gradient
- Local models may not work
- Grain orientation at the cutting edge can have influence



Introduction (GNDs and KAM)

- GNDs are related to strain gradients (for example bending of a single crystal)
- GNDs also depend upon slip plane orientation w.r.to the bending plane of the crystal
- GNDs can be measured by the measuring the misorientation of crystals within a grain (for e.g. from EBSD)

$$\Delta\theta_i^j = \cos^{-1} \left(0.5 \left[\text{tr} \left(\mathbf{G}_j \cdot \mathbf{G}_i^{-1} \right) - 1 \right] \right), \mathbf{G}_i : \text{orientation matrix of pixel } i$$

Kernel Average Misorientation

$$\text{KAM}_i = \frac{1}{N} \sum_{j=1}^N \Delta\theta_i^j, N: \text{neighbours}$$

Grain Reference

Orientation Deviation

$$\text{GROD}_i = \Delta\theta_i^{\text{ref}}$$

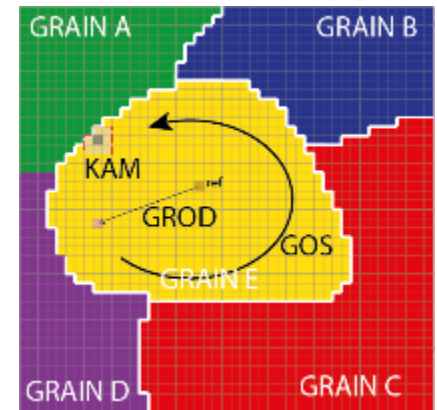
Grain Orientation Spread

$$\text{GOS}_{\text{Grain E}} = \frac{1}{M} \sum_{i=1}^M \Delta\theta_i^{\text{ref}}$$

$$\rho_{\text{GND}} = \frac{2\text{KAM}}{ub}$$

u is step size in EBSD

b is burger's vector magnitude



Strain Gradient Enhanced Crystal Plasticity

- Lower order approach \rightarrow enhanced hardening due to GNDs
- Rate independent crystal plasticity. Slip only on slip planes when resolved shear stress $>$ slip resistance
- Large deformation formulation $\mathbf{F} = \hat{\mathbf{F}}_e \cdot \mathbf{F}_i$ taking $\hat{\mathbf{U}}_e \cong 1$
- Taylor's type hardening $\tau_f^{(\alpha)} = \tau_0 + \mu b \sqrt{\sum_{\beta} Q^{(\alpha\beta)} \rho^{(\beta)}}$

where $Q^{(\alpha\beta)}$ defines strengthening of slip plane α due increase in dislocations density on β

- Dislocation density $\rho^{(\alpha)} = \rho_{SSD}^{(\alpha)} + \rho_{GND}^{(\alpha)}$
- Statistically stored dislocation density

$$\dot{\rho}_{SSD}^{(\alpha)} = \frac{\dot{\gamma}^{(\alpha)}}{\gamma^{\infty}} \left[\rho_{SSD}^{\infty} - \rho_{SSD}^{(\alpha)} \right]$$

$\dot{\gamma}$ is slip rate in slip plane α , γ^{∞} and ρ_{SSD}^{∞} are constants

Strain Gradient Enhanced Crystal Plasticity

- Geometrically necessary dislocation density

$$\rho_{GND}^{(\alpha)} = \frac{1}{b} \sqrt{\rho_{\vdash,GND}^{(\alpha)} + \rho_{\odot,GND}^{(\alpha)}} \quad \begin{array}{l} \rho_{\vdash,GND}^{(\alpha)} \text{ are edge dislocations} \\ \rho_{\odot,GND}^{(\alpha)} \text{ are screw dislocations} \end{array}$$

- Gradient enhanced formulation in lattice configuration

$$\dot{\rho}_{\vdash,GND}^{(\alpha)} = -\mathbf{s}^{(\alpha)} \cdot \nabla^{\#} \dot{\gamma}^{(\alpha)} \text{ and } \dot{\rho}_{\odot,GND}^{(\alpha)} = \mathbf{l}^{(\alpha)} \cdot \nabla^{\#} \dot{\gamma}^{(\alpha)}$$

$\mathbf{l}^{(\alpha)}$ is the unit lattice vector, $\mathbf{s}^{(\alpha)}$ is slip direction and $\nabla^{\#} \dot{\gamma}^{(\alpha)}$ is the gradient of slip rate in lattice configuration

- Gradient enhanced formulation in reference configuration

$$\begin{aligned} \dot{\rho}_{\vdash,GND}^{(\alpha)} &= -\mathbf{s}^{(\alpha)} \cdot \mathbf{F}_i^{-T} \cdot \nabla_0 \dot{\gamma}^{(\alpha)} = -\mathbf{s}_0^{(\alpha)} \cdot \nabla_0 \dot{\gamma}^{(\alpha)} \text{ and} \\ \dot{\rho}_{\odot,GND}^{(\alpha)} &= \mathbf{l}^{(\alpha)} \cdot \mathbf{F}_i^{-T} \cdot \nabla_0 \dot{\gamma}^{(\alpha)} = \mathbf{l}_0^{(\alpha)} \cdot \nabla_0 \dot{\gamma}^{(\alpha)} \end{aligned}$$

- For further details and implementation, see PhD thesis
“Damage in Dual Phase Steels” E.E. Aşık

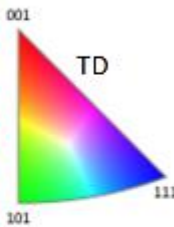
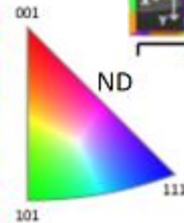
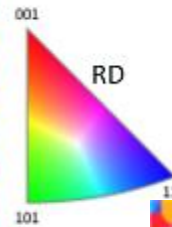
EBSD Measurements

- Oxford Instruments (JSM-7200F, X-MaxN, software: AZtec)
- Undeformed and deformed material
- Sample Preparation

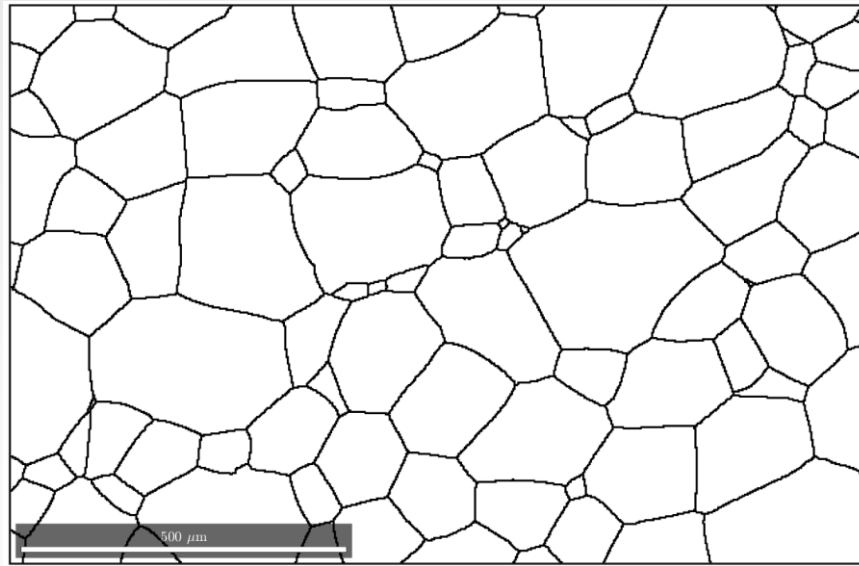
Undeformed material	Deformed material
Embedded in bakelite	Embedded in bakelite
Grinding 2000 paper (3min)	Grinding 2000 paper (10min)
Grinding 4000 paper (3min)	Grinding 4000 paper (10min)
OPS polishing (10min)→check if coating removed →if not keep polishing	OPS polishing (30min)→check if coating removed →if not keep polishing
Rinse with Ethanol and keep overnight in vaccuum	Rinse with Ethanol and keep overnight in vaccuum

EBSD Measurements (Undeformed)

Measurement	Minimum	Maximum	Mean	Standard Deviation
Area (μm^2)	1,609.70	69,838.83	13,772.23	11,552.75
Aspect Ratio	1.11	5.39	1.55	0.46
Breadth (μm)	19.96	270.01	109.22	48.23
Length (μm)	55.46	421.89	160.51	63.66
ECD (μm)	45.27	298.20	122.35	50.70
GOS ($^\circ$)	0.05	5.61	0.44	0.78
Perimeter (μm)	149.71	1,641.57	440.31	200.52



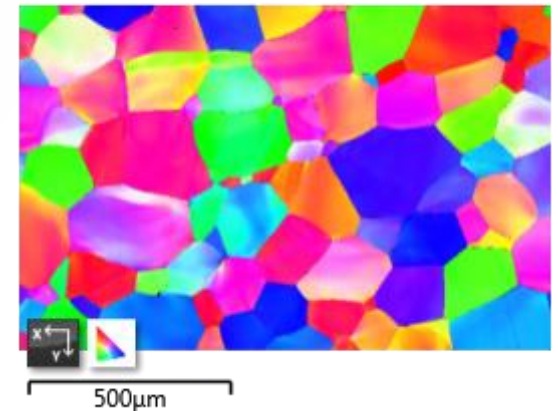
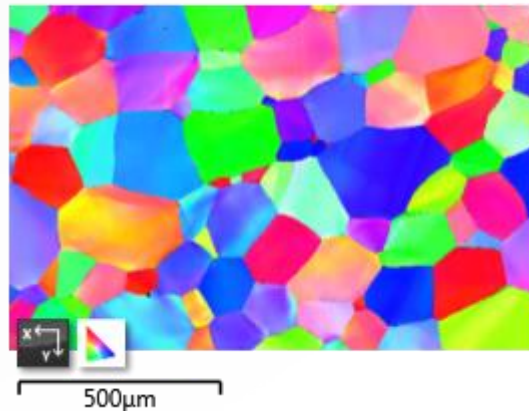
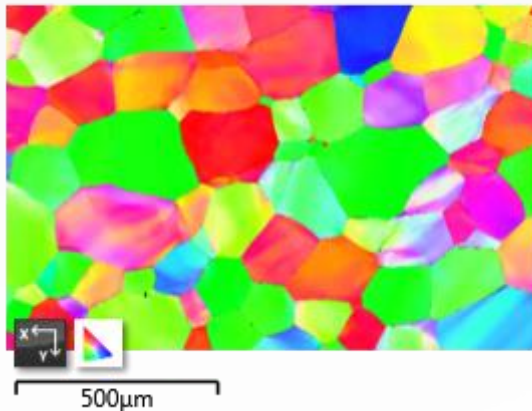
EBSD Measurements (Deformed, rolling)



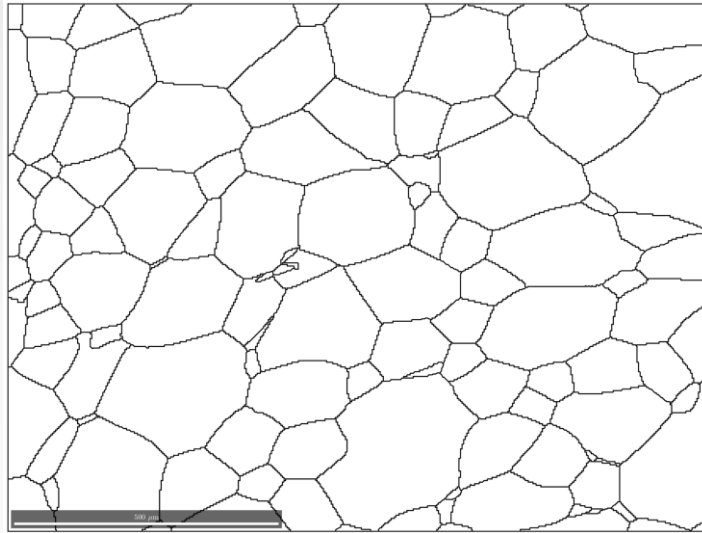
IPF X Color 6

IPF Y Color 6

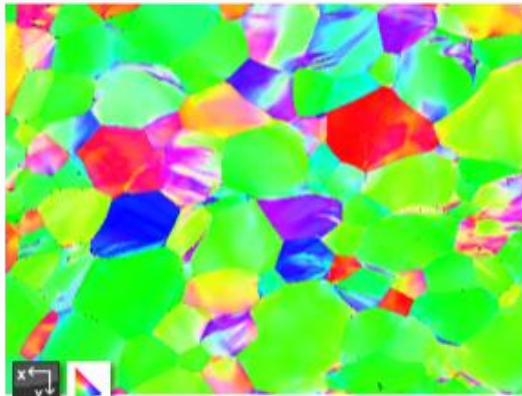
IPF Z Color 6



EBSD Measurements (Deformed, transverse)

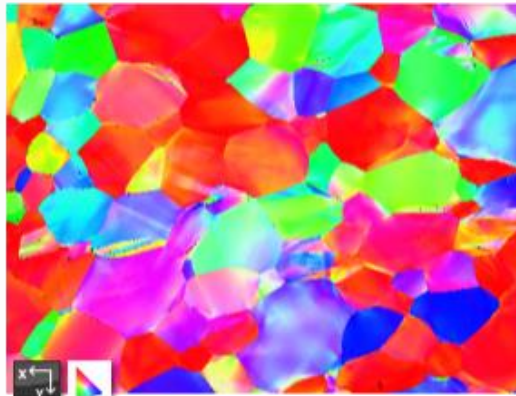


IPF X Color 4



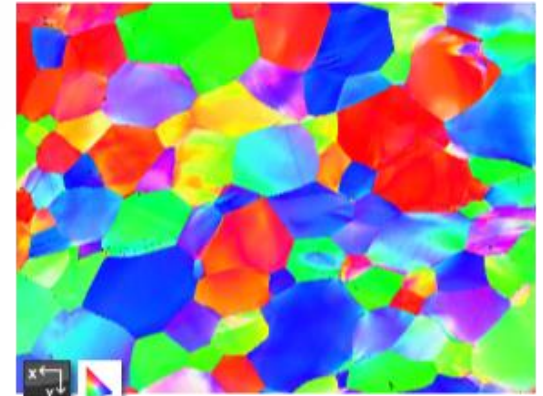
500μm

IPF Y Color 4



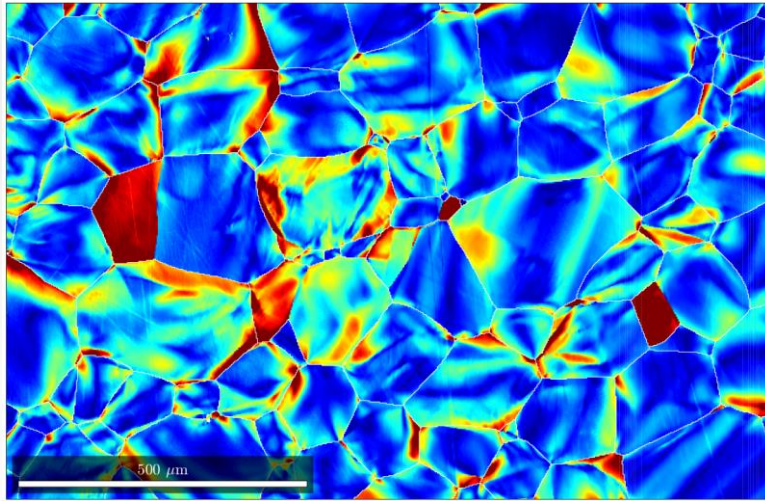
500μm

IPF Z Color 4

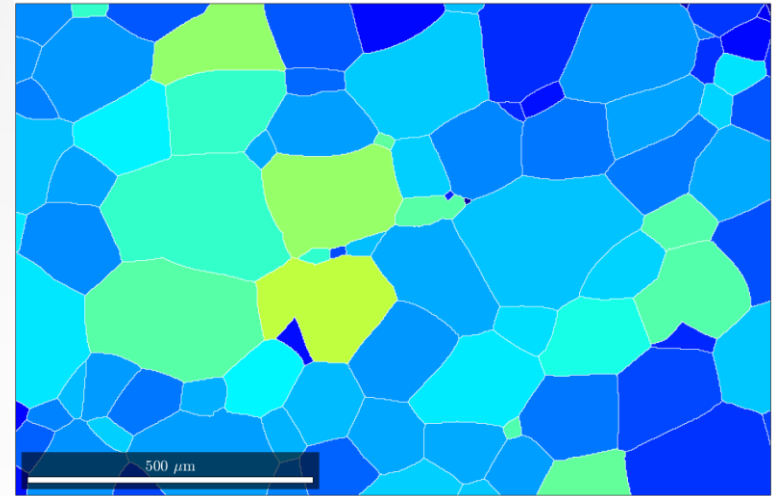


500μm

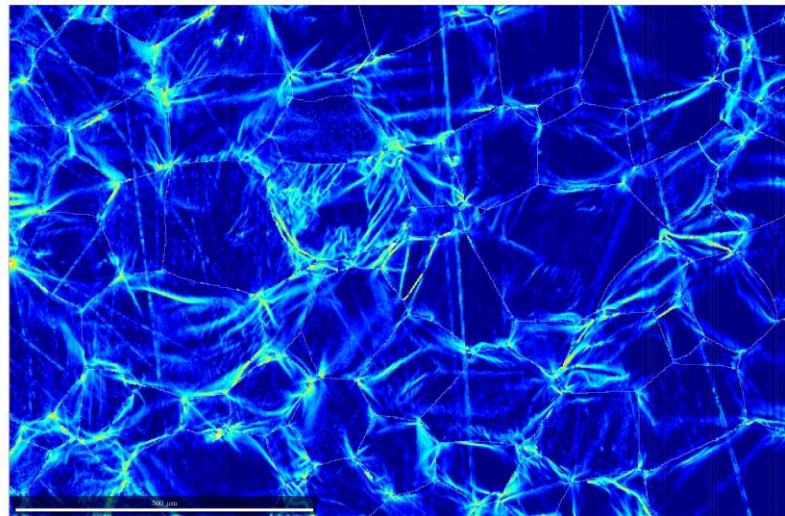
EBSD Measurements (Misorientation)



GROD

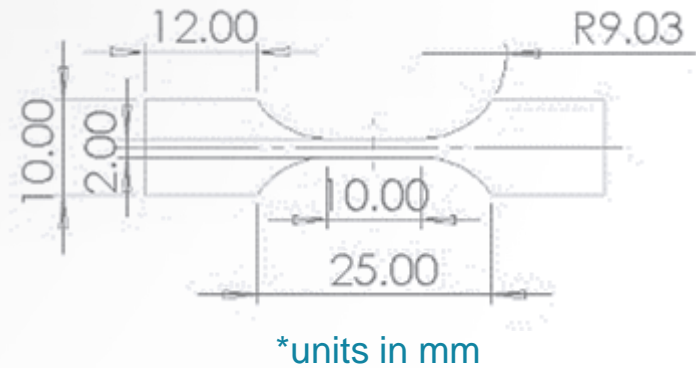
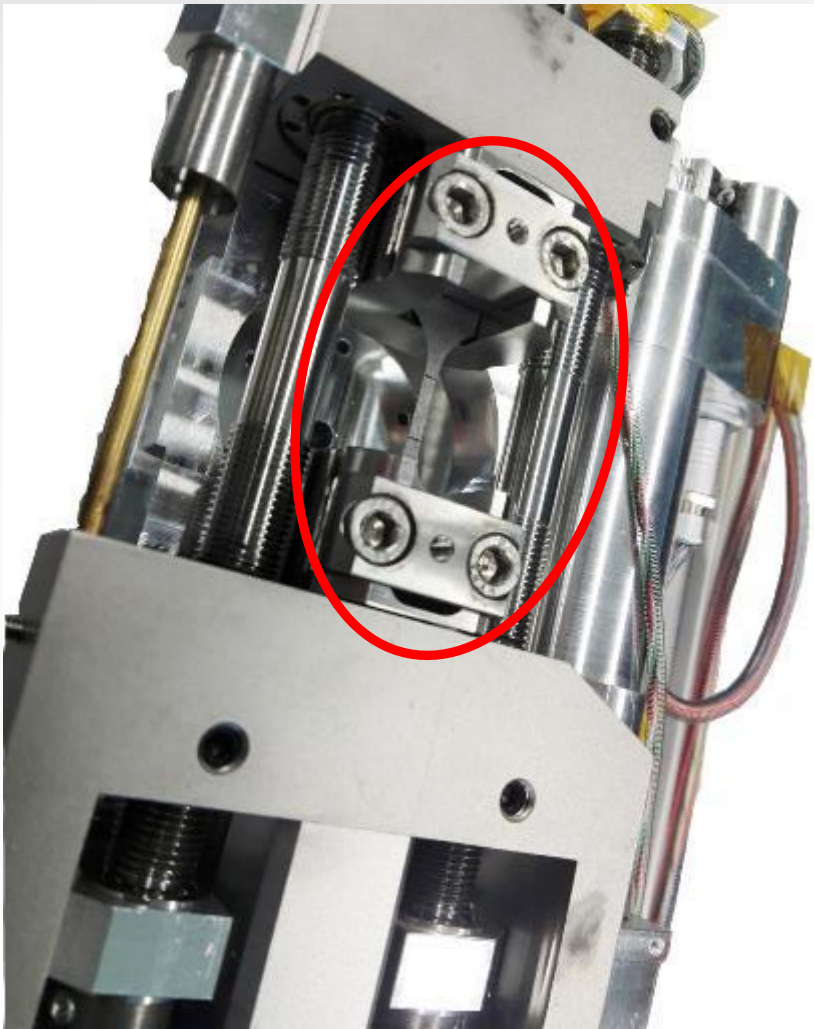


GOS



KAM → GND

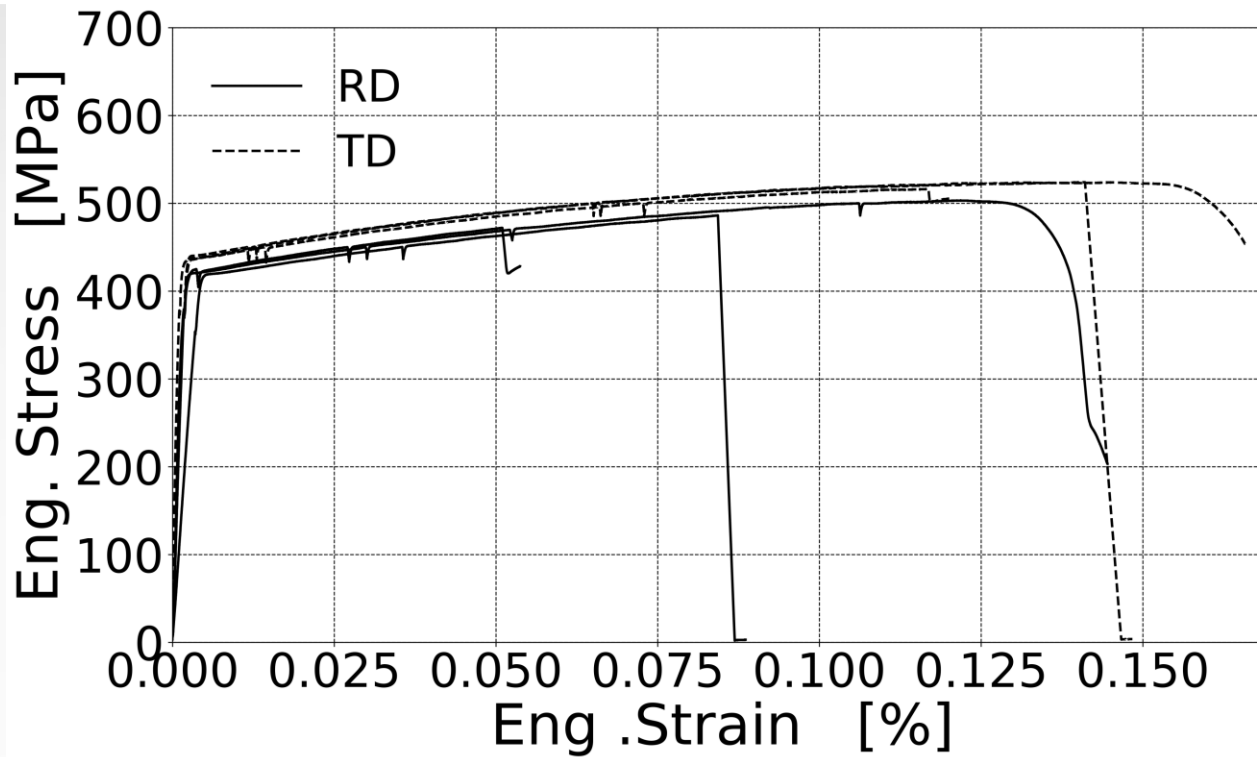
Tensile Tests



R1, R3, R4 → Rolling
T1, T2, T3 → Transverse



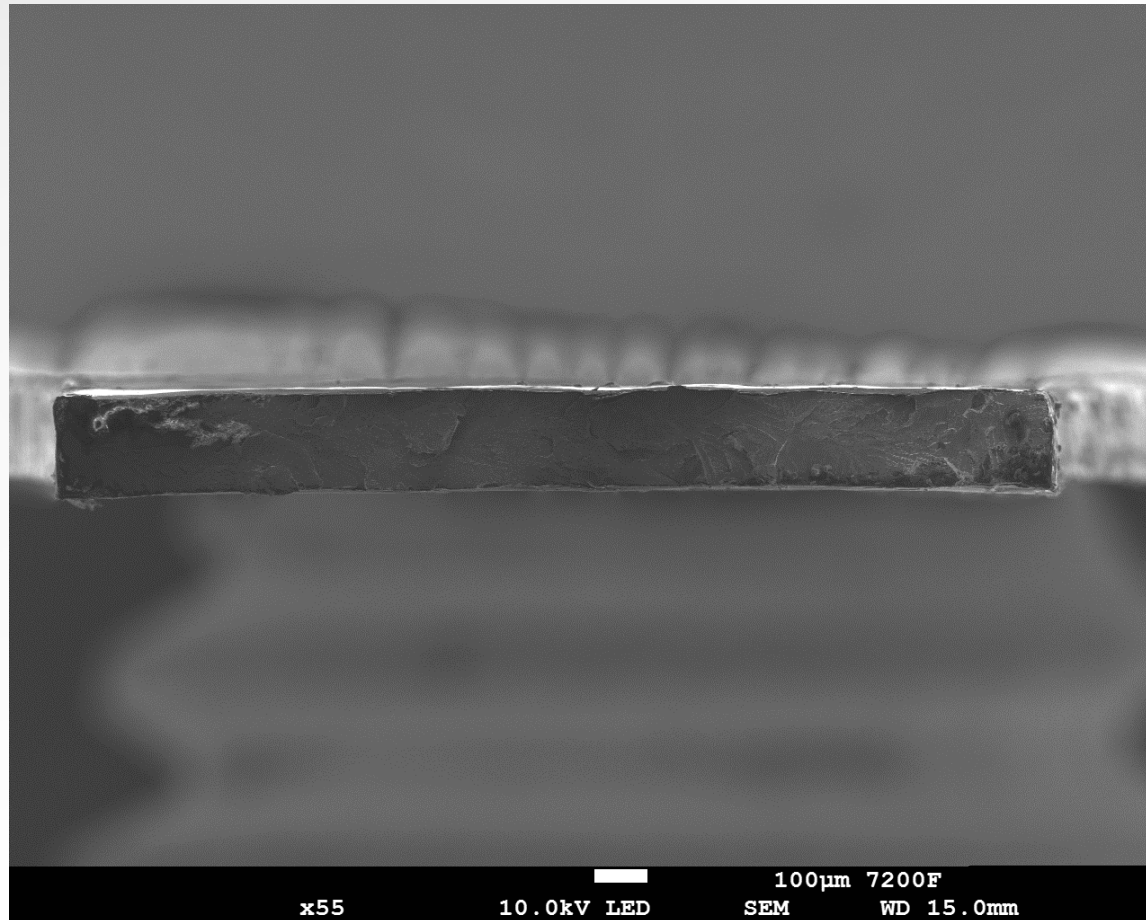
Tensile Tests



	Yield Strength [MPa]	Elastic Modulus [GPa]
Rolling Direction	420 ± 2.8	216 ± 4.0
Transvers Direction	435 ± 2.1	219 ± 1.0

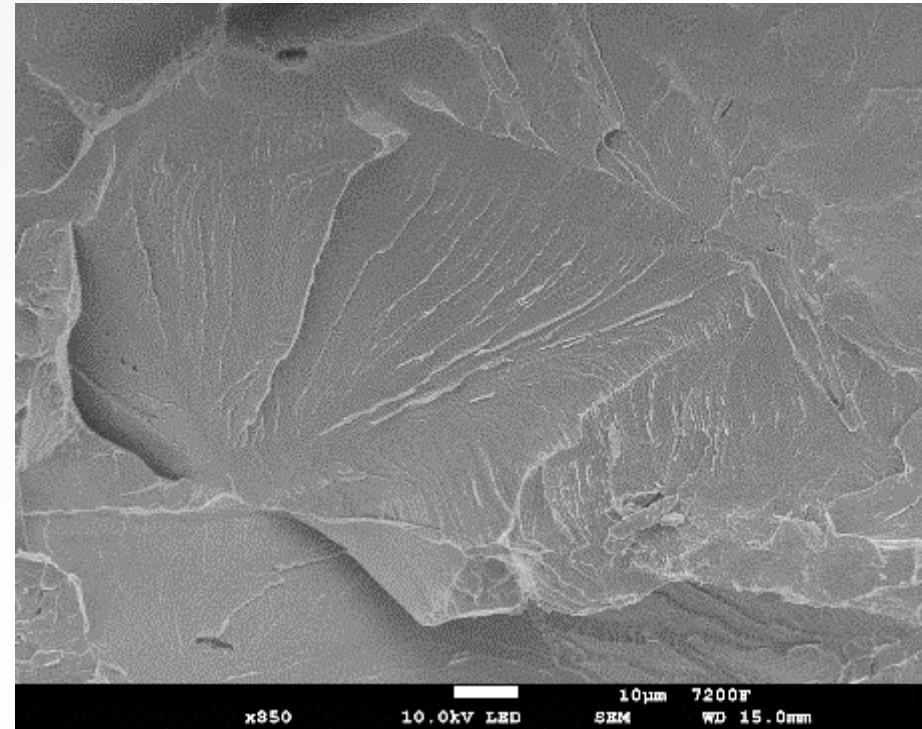
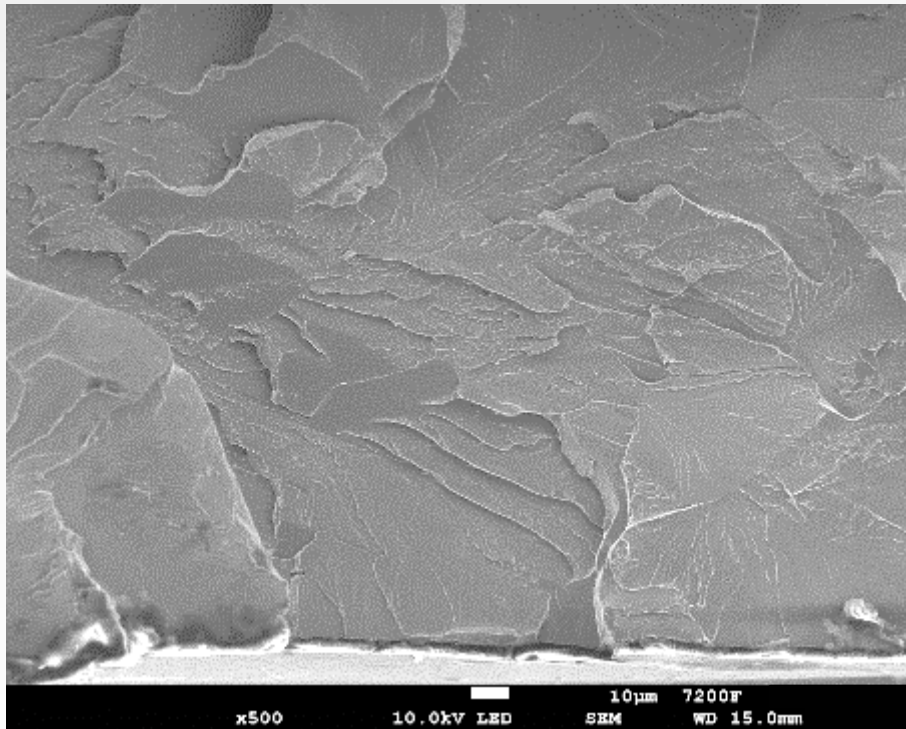
Tensile Tests

Fracture surface : Rolling direction



Tensile Tests

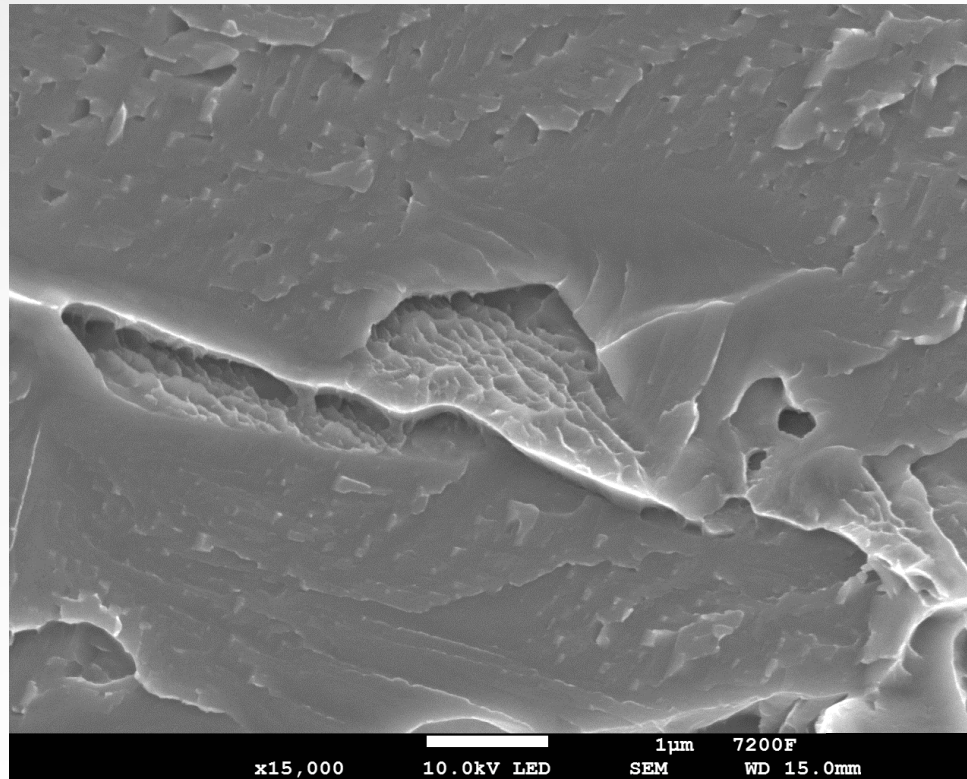
Fracture surface : Rolling direction



Transgranular fracture surface showing river patterns as a sign of brittle fracture behavior.

Tensile Tests

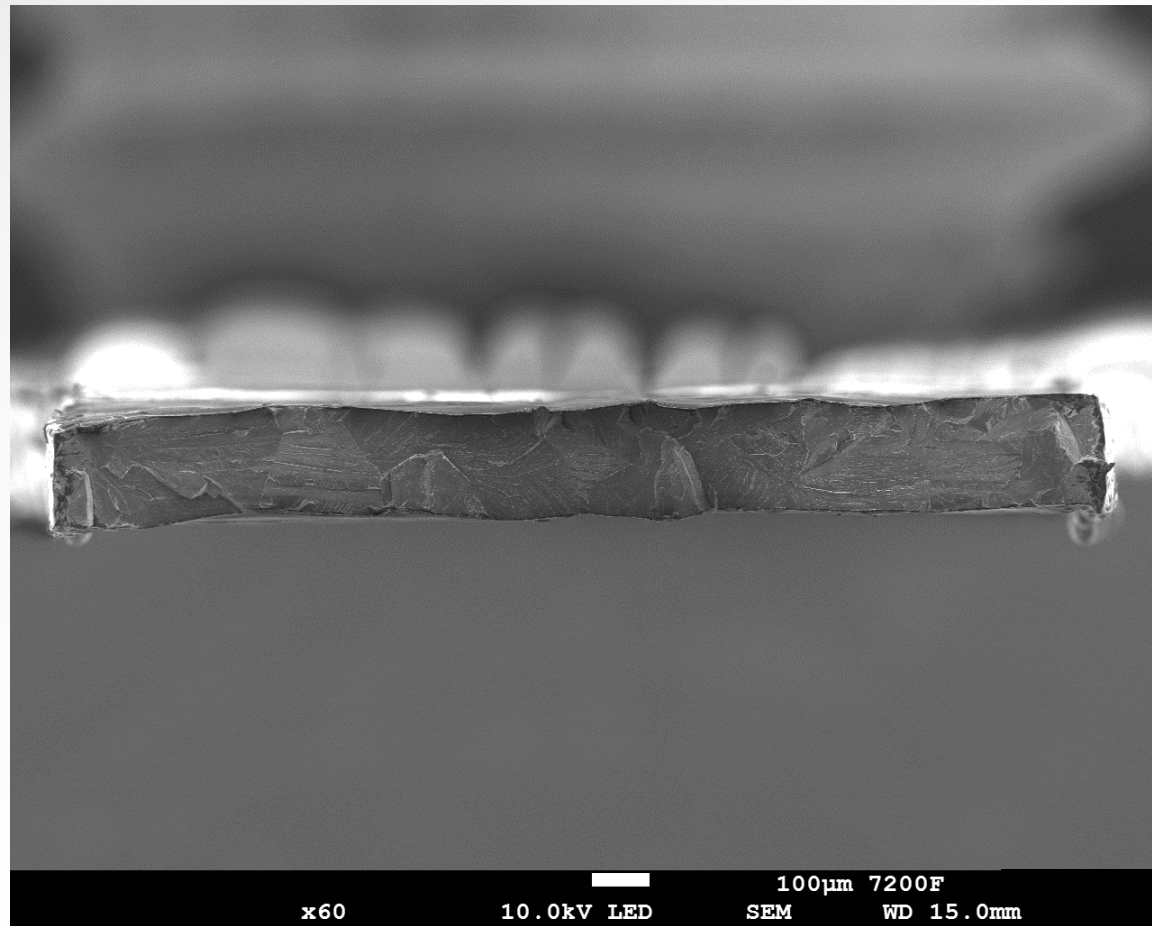
Fracture surface : Rolling direction



Some areas with dimples and micro voids as a sign of ductility.

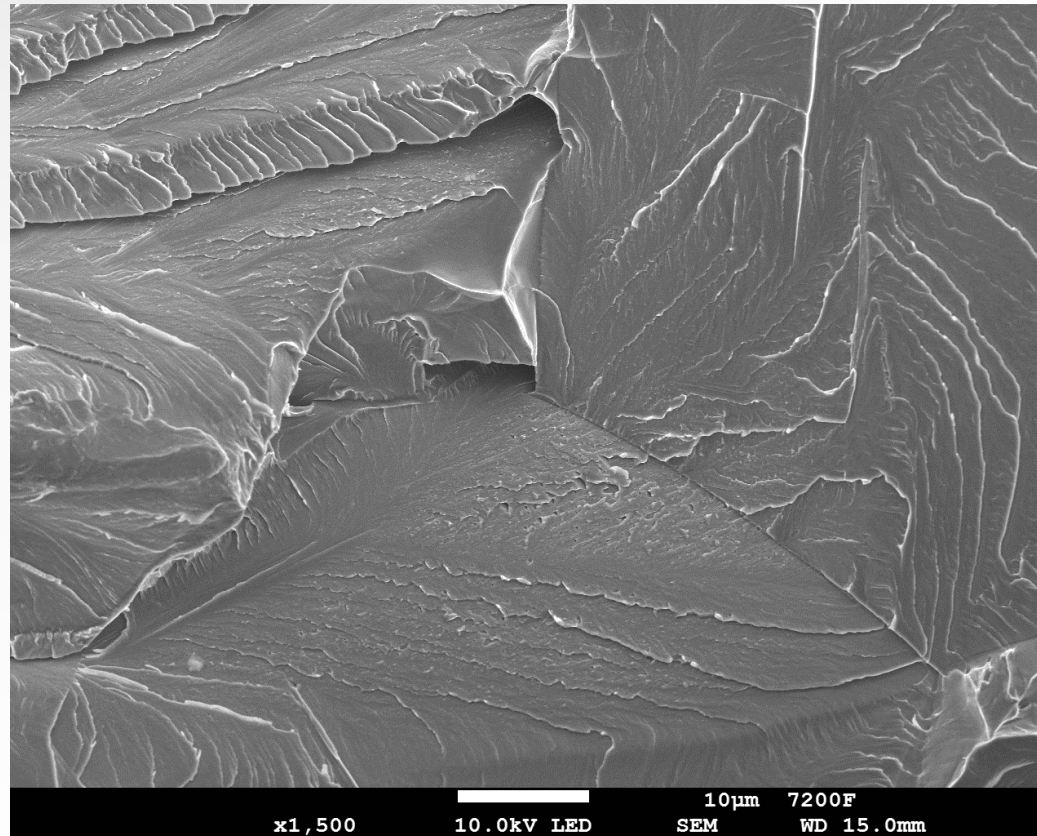
Tensile Tests

Fracture surface : Transverse direction



Tensile Tests

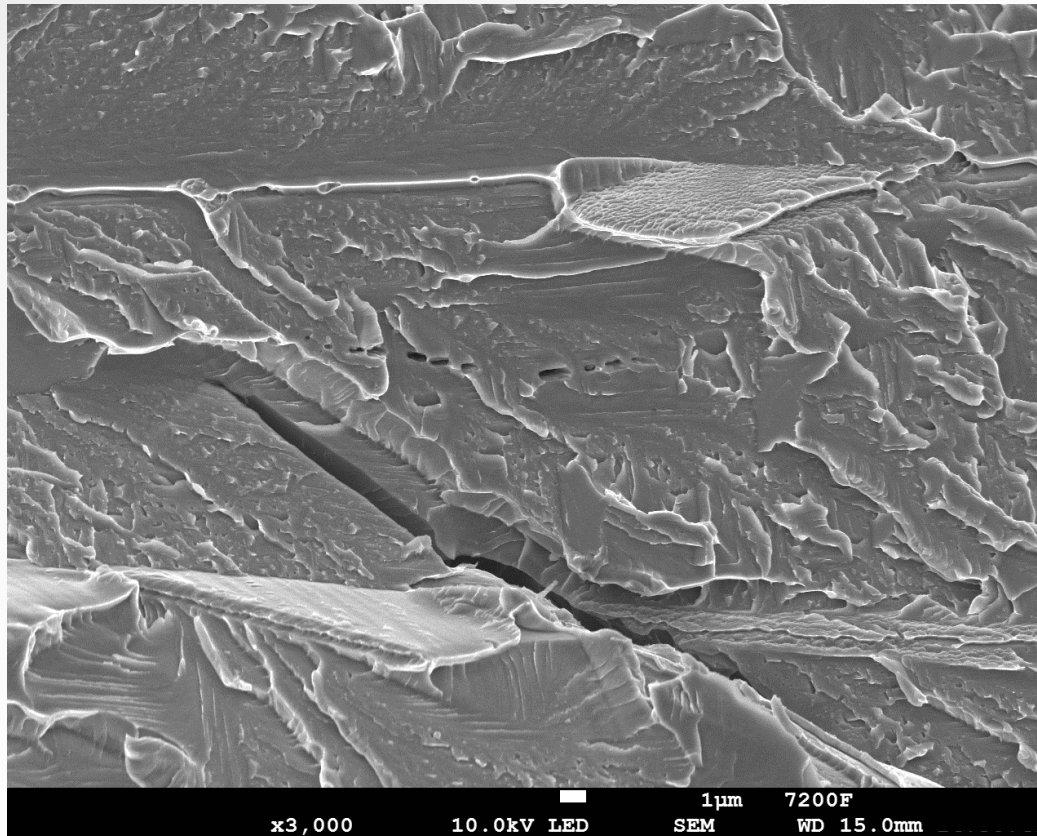
Fracture surface : Transverse direction



River patterns and grain boundary visible.

Tensile Tests

Fracture surface : Transverse direction



Micro voids and dimples

Tensile Test Simulations

- Abaqus Implicit
- User defined subroutines (UMAT, USDFLD)
- Weakly coupled non-local scheme (gradient effect and GND hardening effect in the next increment)
- 2D Plane strain assumption
- Simulation with actual (measured) microstructure and orientations



Tensile Test Simulations

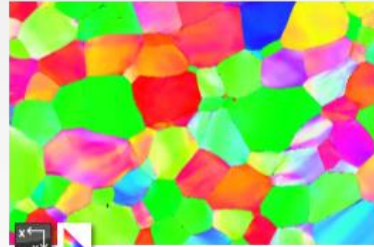
Fully automated

Grain Structure

+

Euler Angles

IPF X Color 6



IPF Y Color 6



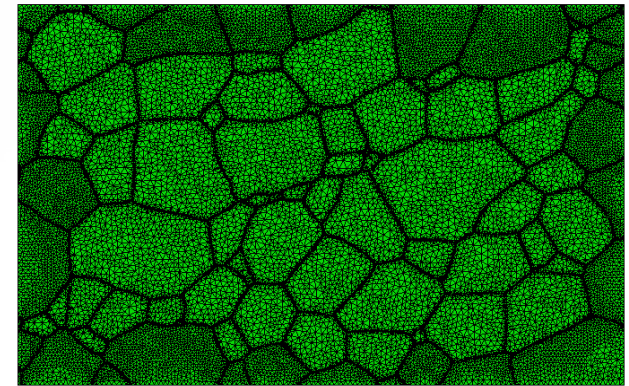
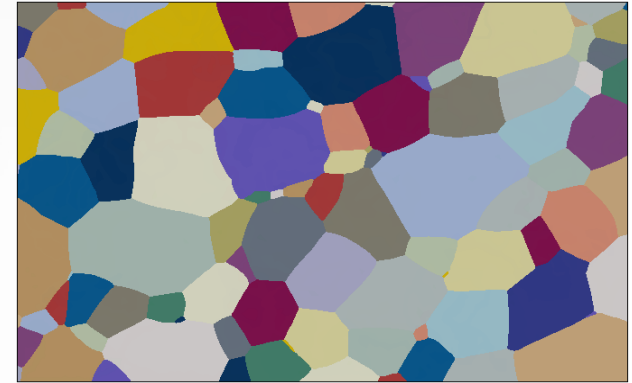
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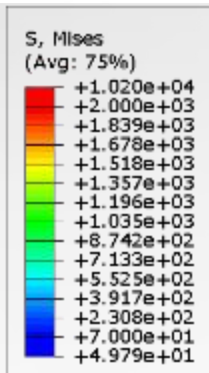
EBSD Measurements

Abaqus Model

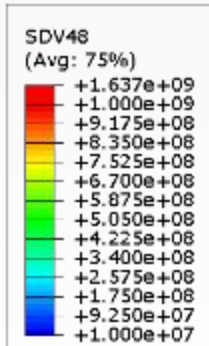
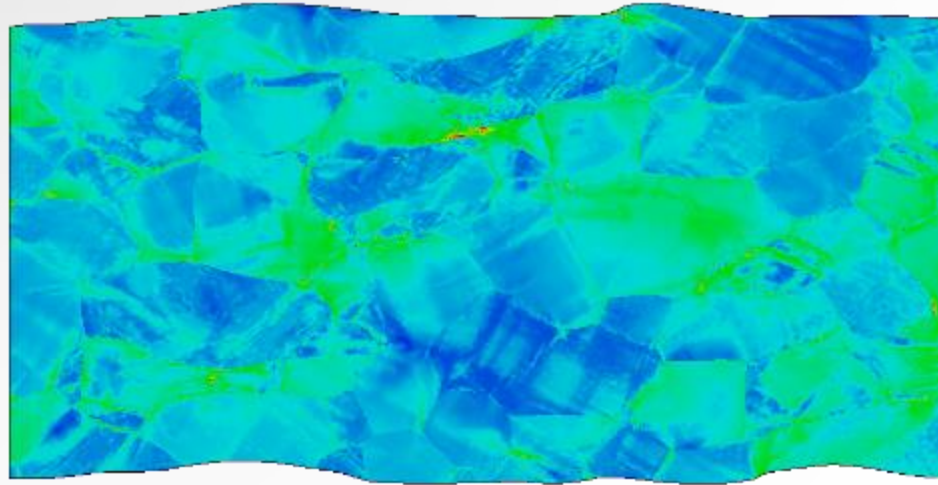
FEA model



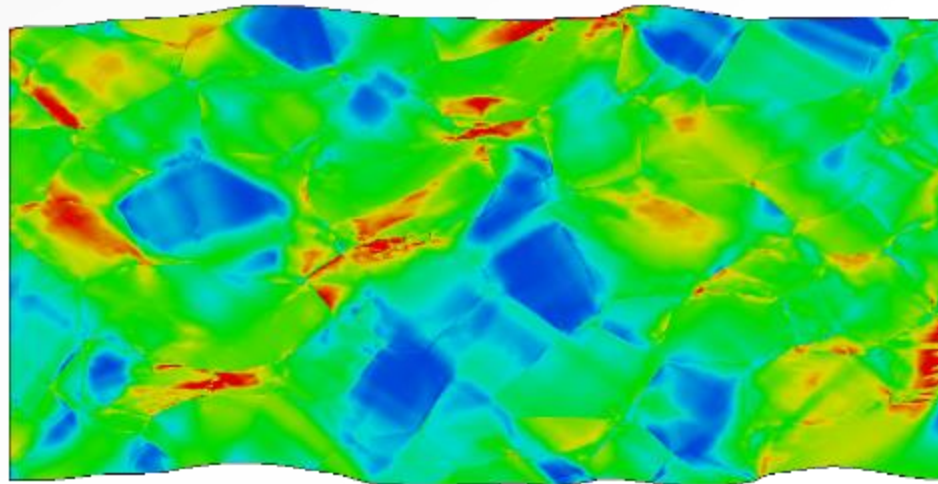
Tensile Test Simulation Results



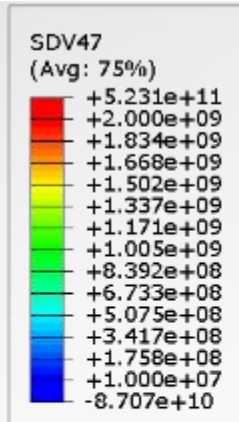
Von Mises Stress



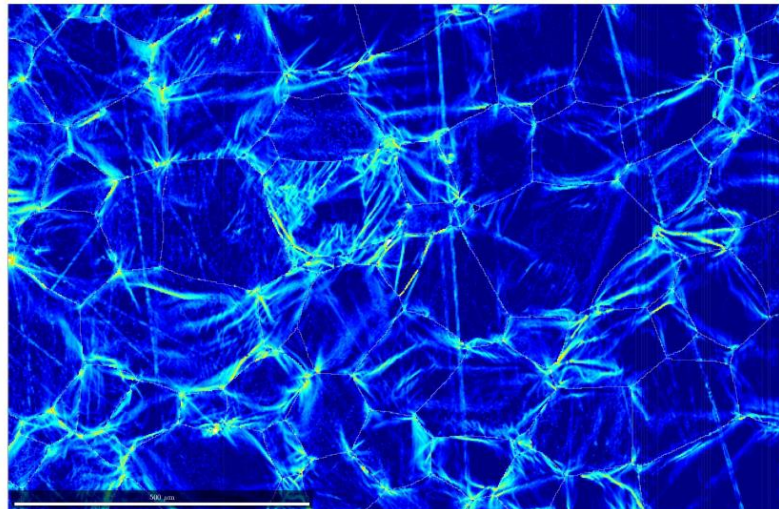
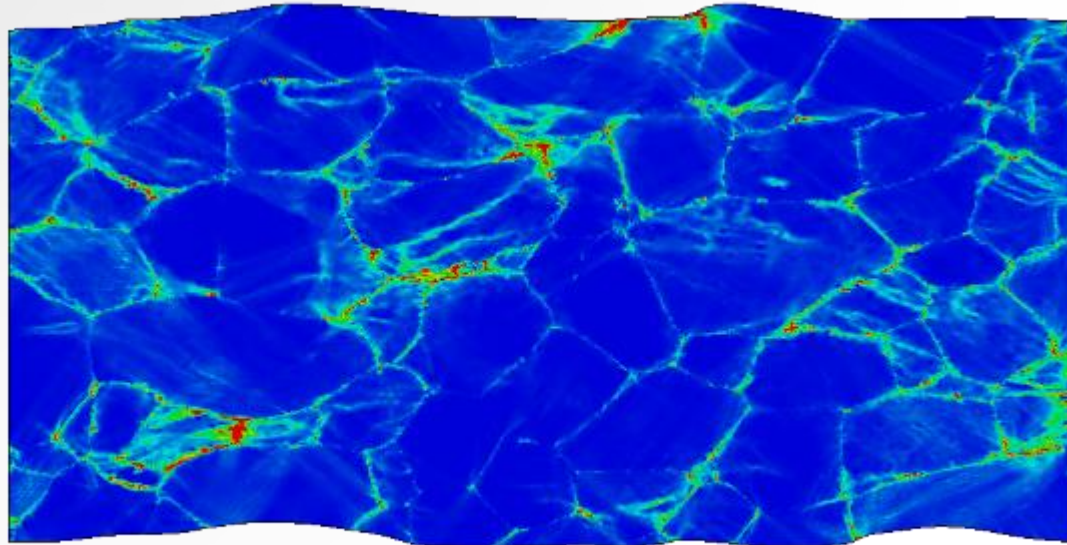
SSD



Tensile Test Simulation results



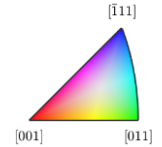
GND density



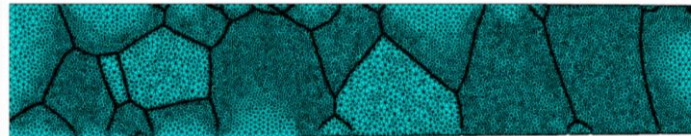
Measured

Tensile Test Simulation results

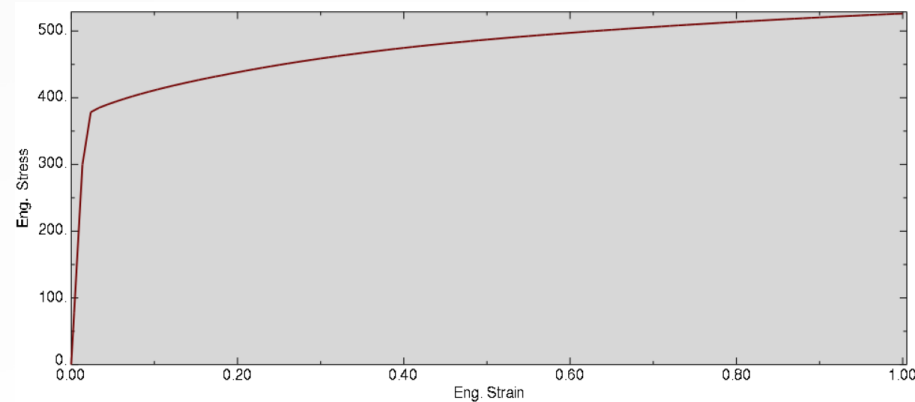
EBSD MEASUREMENT



Abaqus Model



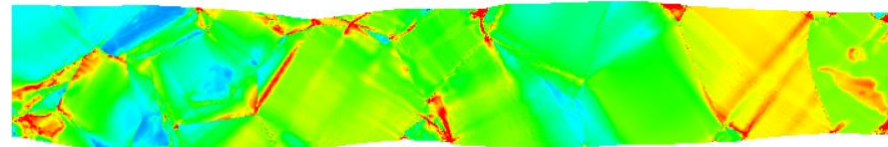
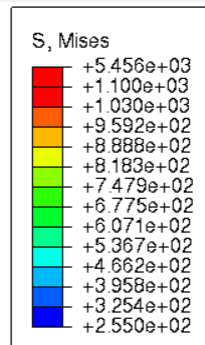
Calibration of material
properties (γ^∞ and ρ_{SSD}^∞)



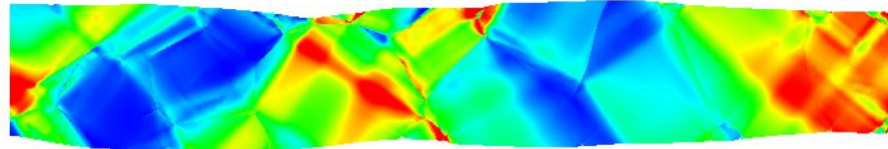
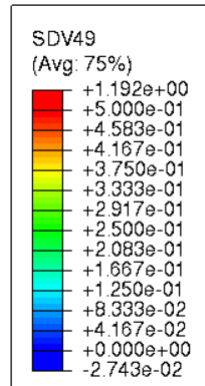
Tensile Test Simulation results



Equivalent
Stress



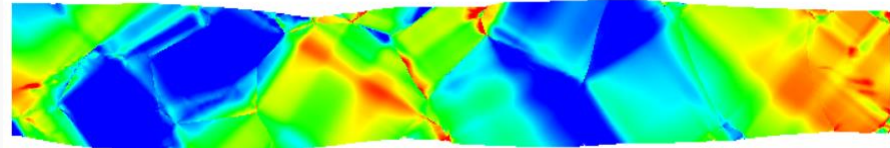
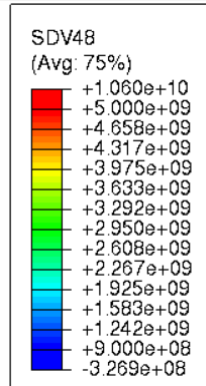
Equivalent
plastic strain



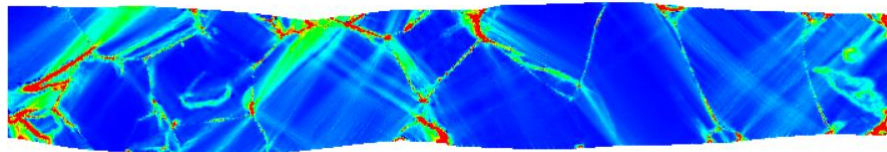
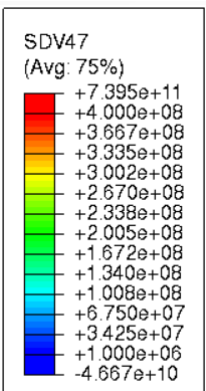
Tensile Test Simulation results



SSDs

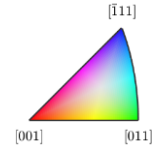


GNDs



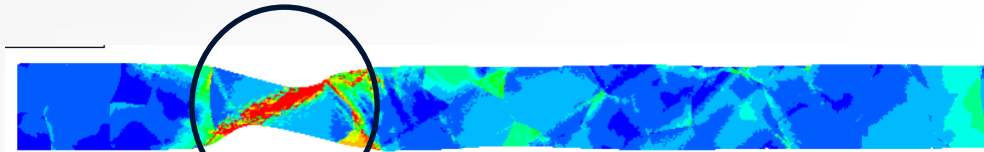
Tensile Test Simulation results

EBSD MEASUREMENT



Abaqus Model

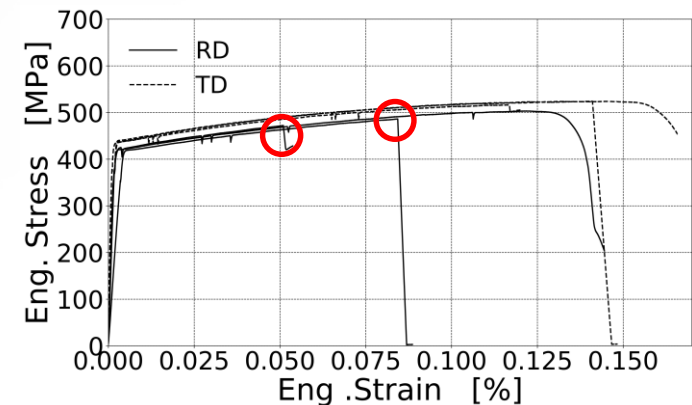
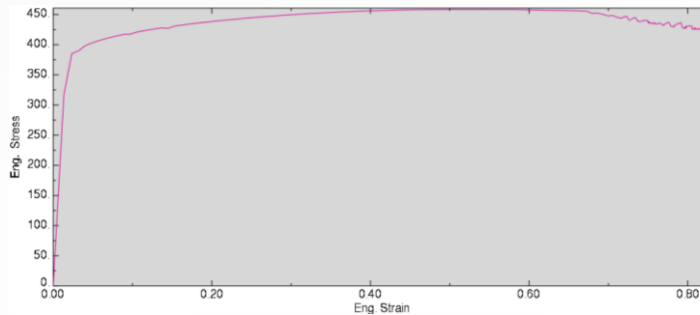
Equivalent Stress



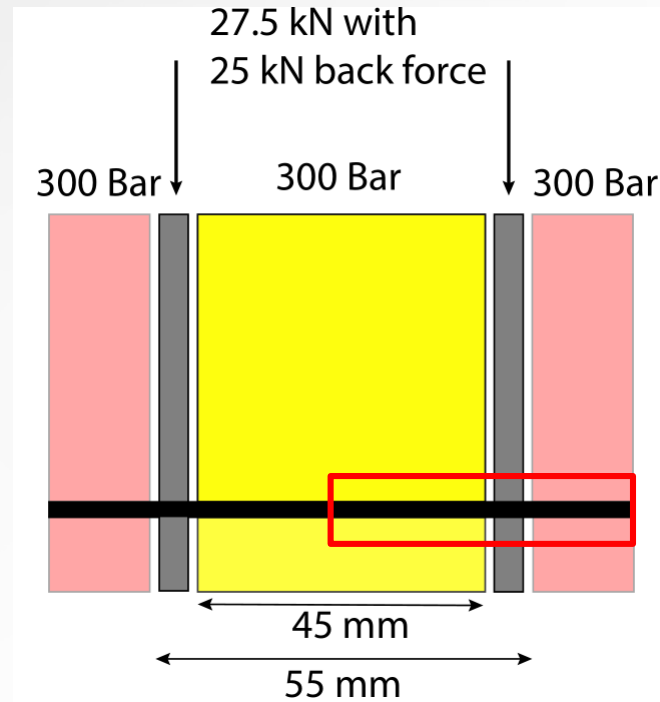
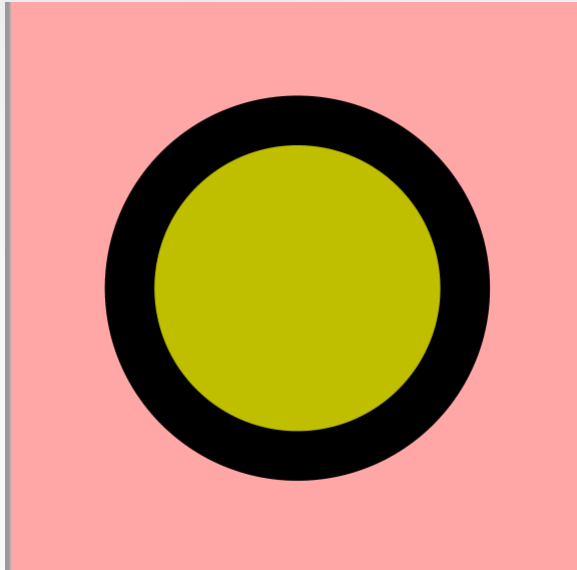
Equivalent plastic strain



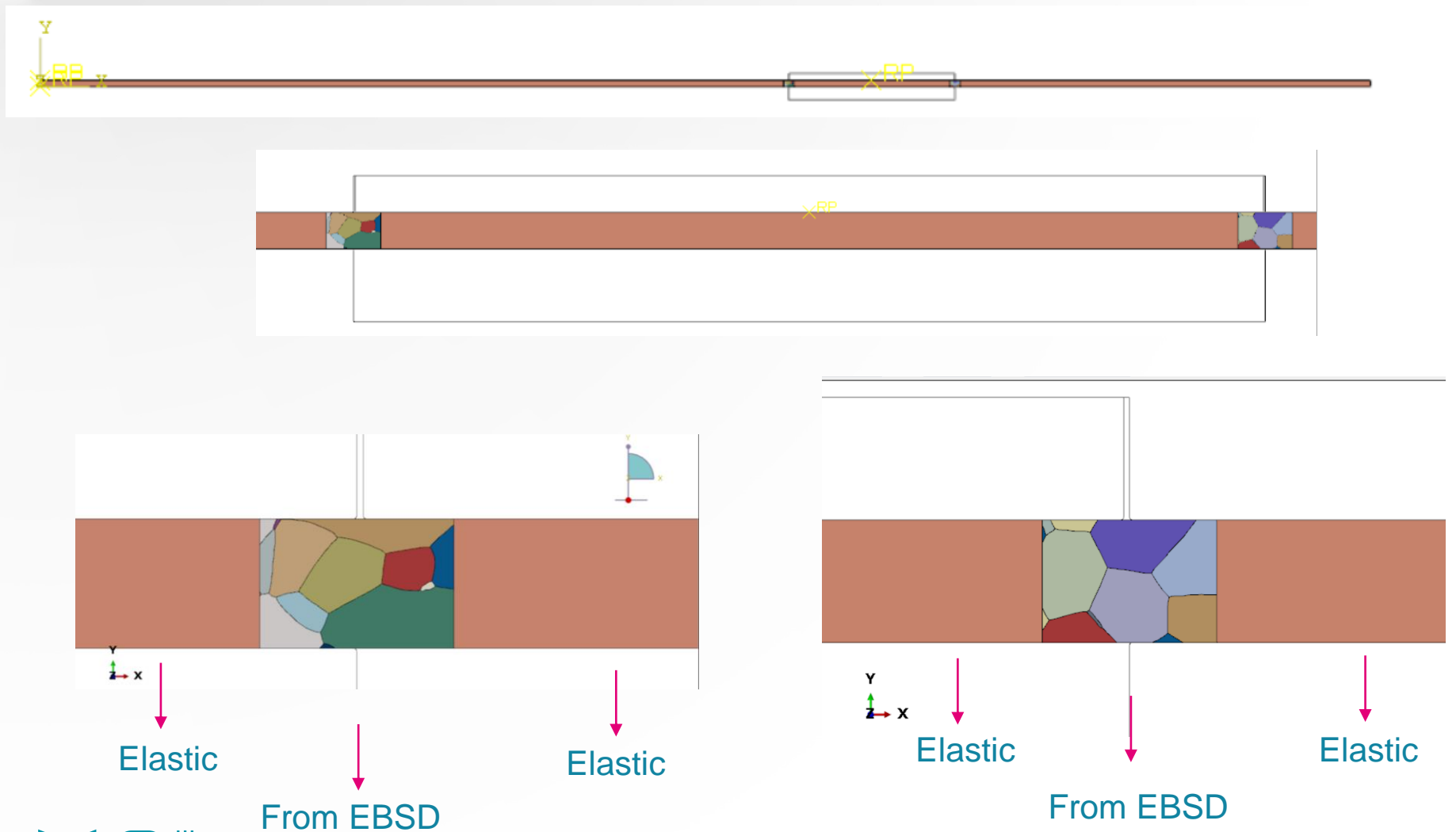
Early failure of samples



Fine Blanking with Crystal Plasticity



Fine Blanking with Crystal Plasticity

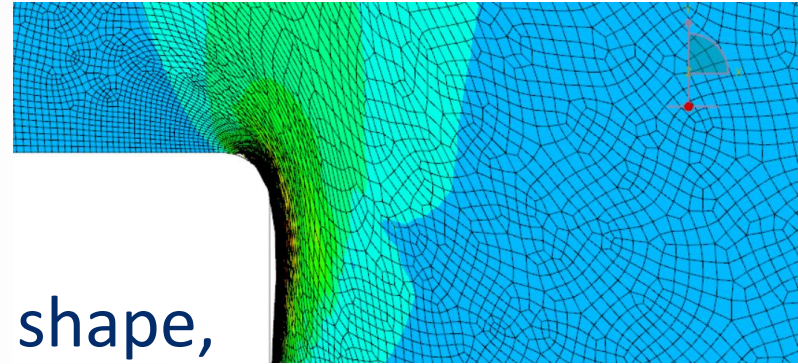


Remeshing Challenges

- Use of UMAT makes remeshing and remapping a challenging task due to tensorial state variables (e.g. orientation remapping)
- Abaqus does not provide an easy solution for remeshing in implicit models

Solution

- Develop remeshing algorithm
- Retain initial (parent) element shape, subdivide and map state variables to child elements
- Subdivision criterion → Improved (child) element shape

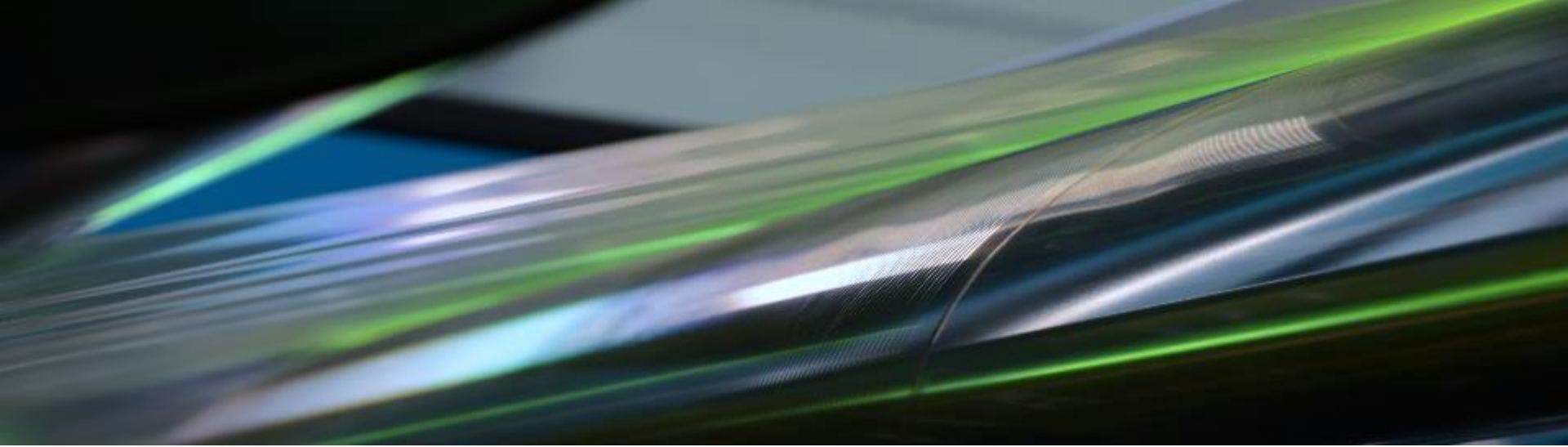


Conclusions

- **Industrial Aim:** Optimize blanking process parameters. The first phase was to simulate the fine blanking process with gradient enhanced crystal plasticity model. Not complete yet, requires a remeshing algorithm.
- The measured grain size, shape and orientations were used in the simulations (to account for variations).
- The material parameters were fitted to the tensile tests.
- Tensile tests show that variation in grain size and orientation can significantly influence the outcome → Requires robust optimization of process parameters.
- The fracture is mainly brittle (transgranular) in nature.

Future Work

- Develop the remeshing algorithm, compatible with gradient enhanced crystal plasticity UMAT.
- Plane strain blanking simulations followed by 3D simulations.
- Simulate with varying microstructure and grain orientations (from measurements)
- Fit the strain gradient enhanced continuum model parameters to 3D simulation results obtained from crystal plasticity simulation.
- Perform robust optimization for the blanking process parameters



THANK YOU!

QUESTIONS???

